In/AuGeNi/Ag/AuM etallization Ohm ic Contacts on n-type GaAs*

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Abstract A new alloy system, $\ln/A u GeN i/A g/A u$, for GaAs ohm ic contact has been proposed and studied in this paper. Our experimental results have shown that this alloy system can not only reduce the ohm ic contact resistivity, but also improve the surface morphology of the contact. The typical contact resistivity of $\ln/A u GeN i/A g/A u$ metallization to n-type GaAs is $5 \times 10^{-7} \Omega \cdot cm^2$. Comparing with AuGeN i/Ag/A u metallization, the new system has a thin In layer inserted between AuGeN i and GaAs. It can overcome so called "balling-up" effect, which offen appears in AuGeN imetallization after annealing. In addition, it was found in our Auger electron spectroscopy profile that Ag can alloy with Au throughout and go into GaAs as well during the annealing at 400. It may be helpful to study of the alloy mechanism.

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1 Introduction

The electrical property and surface morphology of metal-sem iconductor contacts are critical to the performance of sem iconductor devices A s to GaA s devices, such as M ESFET and HEM T, lower source resistance (R_s) is needed to obtain higher transconductance and lower noise figure. In general, ohmic contact resistance takes the most part of R_s . To achieve perfect ohmic contacts, not only low ohmic contact resistance but also reliable thermal stability and smooth surface of metallization after annealing are required

Up to now, various metallization systems for ohmic contacts to GaAs have been studied. A uGeN i system is commonly used to form low resistance ohmic contacts to n-type $\text{GaA} \text{ s}^{[1]}$. It is mainly because the meatl-semiconductor barrier between A uGeN i and n-type

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GaAs is quite low, less than $0.35 \text{eV}^{[2]}$. A uGeN i also has wide process window by various annealing methods

However, "pits" on the surface of contacts are formed on account of the drastic reaction between Au and GaAs And the phenomenon of balling-up is frequently happened due to the low melting point of AuGe and AuGa formed during annealing [4,8]. Generally, in order to overcome this shortcoming, Agor Ti, as a diffusion barrier, is inserted between AuGeN i and upper metal Au to improve on the surface roughness of contacts [1].

Several metallization systems without Au have been investigated. For example, Murakami, etc. have reported the improvement on the surface roughness using N iGe alloyed with n-type GaAs at 600 $^{[3]}$. But the ohmic contact resistance is $1\cdot3\Omega\cdot\text{mm}$, much higher than AuGeN i's. They have also reported that the ohmic contact resistance was reduced after inserting a third metal layer, such as Au, Ag, Pd, In, between Ge and Ni. The lowest contact resistance, $0\cdot28\Omega\cdot\text{mm}$, had been achieved by annealing Ni(60nm)/In(3nm)/Ge (10nm) at 700 for 5 s. This resistance is nearly the same as AuGeN i's. Unfortunately, the high annealing temperature is not suitable to GaAs materials, especially to hetero-epitaxy materials

We fabricated In/A uGeN i/A g/A u ohm ic contacts to seek for lower contact resistance and smooth surface. It is effective to insert a thin In layer between A uGeN i and GaAs for this purpose V arying the thicknesses of In and A uGeN i layers and annealing temperature, we investigated the contact resistance and roughness of surface, Our experimental results showed that the typical contact resistivity decreased to $5 \times 10^{-7} \Omega \cdot \text{cm}^2$, and smooth surface of contacts was realized

2 Fabrication and measurement of ohmic contacts

The material used in this investigation was sem i-insulate (100) GaAs After Si ion implantation and two steps lamp annealing, 450 2s/800 10s, the $2 \times 10^{18} \text{cm}^{-3} \text{ n}^{+}$ layer about 200nm was formed. The sheet resistance of the n^{+} layer was about 170Ω /.

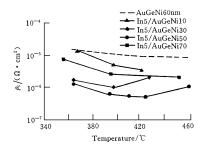
We applied Transmission L ine Method (TLM) to measure the specific ohmic contact resistance^[5,6]. Test patterns were defined as follows At first, mesas were isolated by etching of GaAs wafer using H_3PO_4 H_2O_2 H_2O_3 = 1 100 Secondly, the source and drain windows were opened by photo-lithography followed by removing the remains of photo-resist in windows with oxygen plasma. The oxide layer on surface was removed with NH4OH H_2O_3 = 1 15 for 30 s. Then, In, AuGeN i, Ag, and Au were evaporated in sequence with

 $H_2O = 1$ 15 for 30 s. Then, In, AuGeN i, Ag, and Au were evaporated in sequence with background pressure of 5×10^{-4} Pa. We used the conventional "lift-off" technique to define metal patterns after the evaporation. At last, chips were annealed in a hot plate oven in flow ing N₂ for 50 s.

To acquire the value of resistance accurately, we applied 4 probes method to eliminate the error due to the voltage drop across the probes Auger electron spectroscopy (AES) was used to acquire element composition. The roughness of the surface after annealing was in spected with optical micro scope

3 Results and discussion

Relations of ohm ic contact resistivites of \ln/A uGeN i/A g/A u versus annealing temperatures are shown in Fig. 1 (a) and (b). The dashed lines are for A uGeN i 60nm /A g/A u while solid lines for \ln/A uGeN i/A g/A u. The thicknesses of A g and A u layers, 40nm and 160nm respectively, are the same to all samples A nd annealing times are all 50 seconds



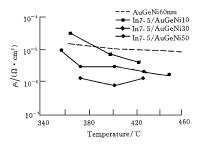


FIG 1 Contact resistivity versus annealing temperature

(a) The thickness of In layer is 5nm

(b) The thickness of In layer is 7.5nm

From Fig. 1, it can be seen that the ohm ic contact resistance of In/A uGeN i/A g/A u metallization is smaller than A uGeN i/A g/A u's The ohm ic contact resistance of the later is about $0.32\Omega \cdot \text{mm}$, the corresponding contact resistivity (ρ_c) is about $8 \times 10^{-6} \Omega \cdot \text{cm}^2$. This result is in the same level of $5 \times 10^{-6} \Omega \cdot \text{cm}^2$ reported before ^[7,8]. But for the former, a typical contact resistivity, $5 \times 10^{-7} \Omega \cdot \text{cm}^2$, was obtained with In (5nm)/A uGeN i (50nm)/A g/A u metallization annealing at 400 . The ohm ic contact resistance is below $0.09\Omega \cdot \text{cm}^2$, which is only 28% of A uGeN i's W hat's more, the surface of the metallization ohm ic contacts with In is much smoother than that without one, as shown in Fig. 2 (a) and (b).

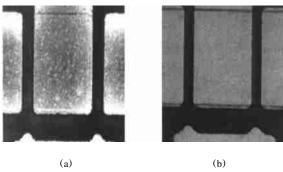
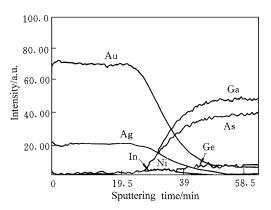


FIG. 2 Photos of metal surface after annealing
(a) Surface of A uGeN i/A g/A u
(b) Surface of In/A uGeN i/A g/A u

Inserting an In layer between AuGeN i and GaAs is the key to reduce the contact resis-

tance There may be two reasons for our results Firstly, In alloyed with GaAs and formed InGaAs at the interface between metal and GaAs. The barrier of alloyed metal and semi-conductor decreased for the existence of InGaAs. So, the contact resistance decreased for tunneling current increased. Secondly, In wetted GaAs well for better adherent ability than AuGeNi Therefore, In inhibited AuGeNi from balling up and a uniform alloyed interface can be obtained, which should be beneficial to get low contact resistance and smooth surface.

The thickness of AuGeN i layer in In/AuGeN i/Ag/Aum etallization is another important factor for the contact resistance As is shown in Fig. 1 (a), the contact resistivity decreases when the thickness of AuGeN i increases from 10nm to 50nm; But the resistivity turns to increase when the thickness increases from 50nm to 70nm. Similar curves can be



seen in Fig. 1 (b). This is because that as the thickness of AuGeN i increases, more Ge atoms diffuse into GaAs as donors resulting in the increment of doping level at the interface. Therefore, the tunneling current can also be enhanced, which makes the contact resistance decrease. However, when AuGeN i is thick enough, much more AuGa is formed for the reaction between Au and GaAs. The barrier between AuGa and GaAs is about

FIG. 3 Profiles of metallization with In after annealing $0.7 \text{eV}^{[2]}$, much higher than the between

A uGeN i and GaA s, 0.35eV. Thus the contact resistance turns to increase To analyze the ohm ic contact further, A ES profile is plotted as Fig. 3 Ag formed solid solution with A u after annealing And Ag diffused into GaA s as well as Au.

4 Conclusion

We have fabricated In/A uGeN i/A g/A u ohm ic contacts and compared it with conventional A uGeN i ohm ic contact. Lower ohm ic contact resistivity, $5 \times 10^{-7} \Omega \cdot \text{cm}^2$, and smoother surface have been achieved after annealing In (5 nm)/A uGeN i (50 nm)/A g (40 nm)/A u (160 nm) at 400 for 50 s Besides, it was found that A g could alloy with A u throughout and go into GaAs as well after annealing at 400. A thin layer of In between A uGeN i and GaAs changed contact resistivity and surface morphology greatly.

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