# A Novel Technology for Post-Treating of Porous Silicon Thick Films in H<sub>2</sub>O<sub>2</sub>\*

Long Yongfu<sup>1, 2</sup>, Zhu Ziqiang<sup>1</sup> and Lai Zongsheng<sup>1</sup>

(1 Department of Electronics, East China Normal University, Shanghai 200062, China) (2 Department of Physics and Electronics, Changde Normal University, Changde 415000, China)

Abstract: The solution of H<sub>2</sub>O<sub>2</sub> is proposed to post-treat thick porous silicon (PS) films. The prepared PS film as the cathode is applied about 10mA/cm<sup>2</sup> current in mixture of ethanol, HF, and H<sub>2</sub>O<sub>2</sub> solutions, which is expected to improve the stability and the smoothness of the surface and the mechanical property of the thick porous silicon films. The microstructure of the PS thick films with thicknesse of 20μm and 70μm has been studied. The SEM images show significant improved smoothness on surface of PS films, and XRD spectra suggest the formation of oxide layer after post-treating in H<sub>2</sub>O<sub>2</sub>.

Key words: porous silicon; post-treatment; H2O2

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### 1 Introduction

Since the photoluminescence properties<sup>[1]</sup> of porous silicon (PS) were discovered in 1990, porous silicon has applied in many fields. During the last decade porous silicon has been a widely studied material due to special properties of porous silicon and its possible applications, such as light emitting devices, sensors, biomedical applications, micro-machining devices as the sacrificial layer in radio frequency (RF)/microwave circuits<sup>[2]</sup> and a thick film as a low-loss or high resistivity interlay-

er<sup>[3]</sup> for passive elements in the monolithic microwave integrated circuit. However, porous silicon, especially, that with high porosities and thick porous silicon films is very unstable in the air. After the formation of a high porous or thick PS films, when the electrolyte and hydrogen evaporate out of the pores in the air, a cracking of the layer occurred. It excessively restricts the applications of porous silicon.

Many methods or technologies such as pentane drying, supercritical drying, freeze drying and slowing evaporation rates have been proposed to posttreat porous silicon films to reduce or eliminate the

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Long Yongfu male, was born in 1965, PhD candidate. His research interests focus on semiconductor materials, the numerical simulation and design of MEMS devices in microwave/RF fields.

Zhu Ziqiang male, was born in 1960, professor and advisor of doctoral students. Currently, his research activities are in MEMS, semiconductor materials and devices.

Lai Zongsheng male, was born in 1943, professor and advisor of doctoral students. Currently, his research activities include MOEMS, microsensors.

capillary stress. Pentane drying is the easiest method to implement. Pentane has a very low surface tension, and shows no chemical interaction with PS. All the above methods or technologies let the electrolyte and the hydrogen evaporate out of the pores of porous silicon films. Because the surface of porous silicon films is very fragile, a cracking of the layer is usually formed.

In this paper, a novel, effective and simple technology for post-treating of thick porous silicon films is proposed. H<sub>2</sub>O<sub>2</sub> is used to post-treat the surface of porous silicon to reduce Si—H bonds in the internal surface of porous silicon, and make the surface of porous silicon films to form a layer of porous SiO<sub>2</sub>. The experimental results show that the method can enhance the stability and the smoothness of the surface and the mechanical intensity of the thick porous silicon films in the air.

## 2 Experiment

After the formation of porous silicon, the internal surface of the PS films is very large. The ratio [4] of the surface area over the volume is as high as  $1000 \text{m}^2/\text{cm}^3$ . Such a large surface contains an enormous quantity of impurities coming from the electrolyte used for electrochemical etching and from the ambient air. The main original impurity what is always found in PS films is hydrogen. Infrared absorption (IR) experiments have shown the presence of Si-H<sub>x</sub> groups (x = 1, 2, 3) on the internal PS surface during the etching process<sup>[5]</sup>. In freshly porous silicon samples, there are a lot of Si-H bonds in the internal surface of porous silicon, the atomic ratio [4] H/Si in PS is as high as 0.1 ~ 0.6, depending on the porosity and surface area of the samples. This result means that the surface of freshly etched porous silicon is almost totally covered by Si-H<sub>x</sub> groups. Si-H<sub>x</sub> groups are unstable in the air, and hydrogen will be released, resulting in an unstable surface of porous silicon.

After the formation of porous silicon, a gas/ liquid interface forms inside the pores and a pressure drop<sup>[6]</sup>  $\Delta p$  occurs, during evaporation, which is given by

$$\Delta p = \mathcal{Y} \frac{S \cos \theta}{P}$$

where  $\mathcal{Y}$  is the liquid surface tension, S the interface area,  $\cos\theta$  the curvature of the gas/liquid interface, and P is the porosity.  $\Delta p$  is as high as some MPa.

In order to reduce Si-H<sub>x</sub> groups, we propose that the Si—H bonds are progressively replaced by Si—O bonds through reaction with H<sub>2</sub>O<sub>2</sub> in mixture of ethanol, HF (47%) and H<sub>2</sub>O<sub>2</sub> solutions after the formation of porous silicon.

The chemical reaction in the electrolyte can be described.

$$SiH_x + H_2O_2 \longrightarrow SiO_2 + H_2O + H_2 \uparrow$$

Figure 1 shows the schematic diagram that  $H_2O_2$  attacks the Si—H bonds. After the chemical reaction, the Si—H bonds are replaced by the Si—O bonds. When drying porous silicon films with post-treatment in  $H_2O_2$ , there are not nearly Si—H bonds in the internal surface of porous silicon. During the evaporation, there is not almost hydrogen. A gas/liquid interface forms inside the pores and a pressure drop  $\Delta p$  is very small. Meanwhile, the surface of porous silicon films is protected by the porous SiO<sub>2</sub>. Therefore, there is not a cracking of the layer on the surface of porous silicon films.

Fig. 1 Schematic diagram of H<sub>2</sub>O<sub>2</sub> attacking Si—H bonds

Experimentally, porous silicon was traditionally prepared by anodic electro-chemical etching. The silicon sample and platinum electrode were immersed in an electrolyte which is commonly a mixture of ethanol and HF (47%) (a typical volume ratio is 1 : 1). PS samples were produced by anodization of (100) P-type silicon wafers with  $0.01\Omega$  • cm resistivity with a current density of  $20\text{mA/cm}^2$  in darkness for 21 and 75min, respec-

tively. Thicknesses of porous silicon films were about  $20\mu m$  and  $70\mu m$ , respectively.

After the formation of porous silicon, the porous silicon sample acts as the cathode while the anode was platinum. The porous silicon sample as the cathode was applied about  $10\text{mA/cm}^2$  current in mixture of ethanol (100%), HF (47%) and H<sub>2</sub>O<sub>2</sub> solutions (a typical volume ratio was 10:10:1, respectively) for about 5min. Hydrogen in the cathode will be taken off. At last the porous silicon sample was put into 1% H<sub>2</sub>O<sub>2</sub> to clean for about 20min. Experimental results show that the microconstruction of PS films depends on the volume ratio of H<sub>2</sub>O<sub>2</sub>/HF solution and the time of post-treatment.

#### 3 Results and discussion

In order to evaluate the improved performance by post-treating porous silicon films using H<sub>2</sub>O<sub>2</sub> solutions, four porous silicon samples with thicknesses of 20µm and 70µm were prepared. Two porous silicon samples directly dried in the air after the formation of porous silicon films and the other were post-treated in H<sub>2</sub>O<sub>2</sub> solutions. Figures 2 and 3 show the SEM images on surface of different porous silicon films. Figures 4 and 5 show XRD spectra of thick porous silicon films.

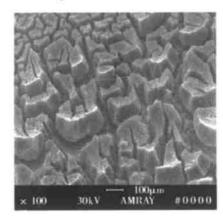


Fig. 2 SEM image of PS(thickness of 70µm) surface without post-treatment (about 67% porosity)

Figure 2 shows a typical example of cracking pattern without any post-treatment. When the elec-

trolyte and hydrogen evaporate out of the pores, the surface of porous silicon will crack. A cracking of the layer is systematically observed. Inter-distance of cracking was about  $1\sim 100\mu m$ .

Figure 3 shows SEM images of porous silicon with post-treatment in H<sub>2</sub>O<sub>2</sub> solutions. After post-treatment, the Si-H<sub>x</sub> groups almost do not present on the inner surface of porous silicon, and the surface is protected by the porous SiO<sub>2</sub> layer. When the electrolyte evaporates out of the pores, the surface of porous silicon will not crack, the micro inter-distance is about 20nm on surface of the porous silicon films.

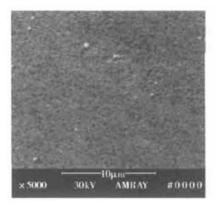


Fig. 3 SEM image of PS(thickness of 70μm) surface with H<sub>2</sub>O<sub>2</sub> post-treatment (about 67% porosity)

Figure 4 shows that XRD spectra of P(100) silicon wafer, PS without any post-treatment and with post-treating in H2O2. XRD spectra of the P (100) type silicon (0.01Ω • cm) wafer in Fig. 4(a) shows a very narrow width and a very sharp peak at 68.783°, which shows the nature of the silicon wafer. XRD spectra are distinct variations between the PS surface without any post-treatment and with post-treatment in H2O2. The PS without any post-treatment shows a broadened small peak at 68.781°. The lattice parameter expansion  $\Delta\alpha/\alpha$  is about 2. 55×10<sup>-5</sup>. The origin of this expansion<sup>[7]</sup> is related to the silicon-hydrogen bonds at the inner surface of the PS films. A very sharp peak in Fig. 4 (c) at 68.874° shows the PS characteristic with post-treatment in H<sub>2</sub>O<sub>2</sub>, and the lattice parameter contraction  $\Delta \alpha / \alpha$  is about 1.27  $\times$  10<sup>-5</sup>. What the formation of oxide layer displaces the Si—H bonds originates a contraction of  $\Delta\alpha/\alpha$  (3.82×10<sup>-5</sup>) due to post-treatment. Moreover, this peak becomes very narrow as shown in Fig. 4(c), which confirms the formation of pores on the surface of porous silicon and there are few Si-H<sub>x</sub> groups in the internal surface of PS films. The crystallite size of PS with post-treatment in H<sub>2</sub>O<sub>2</sub>, which is the representation of the remnant silicon portion after the post-treatment and formation of pores become smaller than which of PS without any post-treatment.

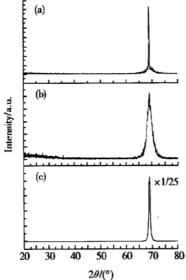


Fig. 4 XRD spectra of P(100) silicon (0.01 $\Omega$  • cm) wafer (a), PS with thickness of 20 $\mu$ m without post-treatment (b) and with post-treating in H<sub>2</sub>O<sub>2</sub>(c)

In Fig. 5 a small peak at 32.62° was observed and the lattice parameter (a) equals 0.272nm, which is the consequence of the formation of SiO<sub>2</sub> (110 $\beta$ ) films after post-treating in H<sub>2</sub>O<sub>2</sub>. The result demonstrates the formation of oxide films due to post-treating in H<sub>2</sub>O<sub>2</sub>. Further, the presence of this peak<sup>[5]</sup> in the PS structures confirms that the cubic structure of the PS is retained after post-treatment.

#### 4 Conclusion

We have presented the solution of H<sub>2</sub>O<sub>2</sub> to post-treat thick porous silicon films (thickness is

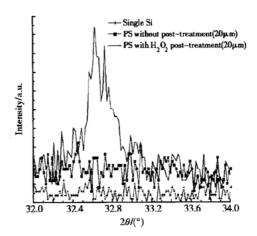


Fig. 5 XRD spectra of different wafers at small  $2\theta$ 

about 70µm), the micro inter-distance of porous silicon is about 20nm on the surface of the porous silicon films. The SEM image shows significant improved smoothness on surface of porous silicon films after post-treating in H<sub>2</sub>O<sub>2</sub>. XRD spectra reveals the formation of oxide films after post-treating PS in H<sub>2</sub>O<sub>2</sub>. This treatment will also enhance the stability, the smoothness and the mechanical intensity of the thick porous silicon films.

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## 用过氧化氢后处理多孔硅厚膜的一种新技术\*

龙永福1,2 朱自强1 赖宗声1

(1 华东师范大学电子系, 上海 200062) (2 常德师范学院物电系, 常德 415000)

摘要:提出使用过氧化氢后处理多孔硅厚膜.在乙醇、氢氟酸、过氧化氢溶液中,多孔硅样片做阴极施加电流密度为 10mA/cm²,希望通过后处理增强多孔硅表面的稳定性、光滑度和机械强度.研究了厚度为 20μm 和 70μm 的多孔硅厚膜经过过氧化氢处理后的微结构.扫描电镜图显示经过过氧化氢处理后的多孔硅厚膜表面的光滑度有极大的提高, X 光衍射光谱揭示经过过氧化氢后处理后多孔硅表面形成了一层氧化膜.

关键词: 多孔硅; 后处理; 过氧化氢

PACC: 3220R; 6146

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龙永福 男,1965年出生,博士研究生,主要从事半导体材料、数值模拟与微波/射频 MEMS器件设计.

朱自强 男, 1960年出生, 教授, 博士生导师, 主要从事 MEMS、半导体材料与器件的研究.

赖宗声 男,1943年出生,教授,博士生导师,目前研究领域集中在微光电系统、微型传感器.